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Structure, Stability, and Electronic Properties of Niobium-Germanium and Tantalum-Germanium Clusters

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Abstract: The structural, electronic and magnetic properties of niobium and tantalum doped germanium clusters $M\text{Ge}_n$ ($M = \text{Nb}, \text{Ta}$ and $n = 1-19$) are investigated by first principles calculations within the Density Functional Theory approach (DFT). The growth pattern behaviors, stabilities, and electronic properties are presented and discussed. Endohedral cage-like structures in which the metal atom is encapsulated are favored for $n \geq 10$. The doping metal atom contributes largely to strengthen the stability of the germanium cage-like structures with the binding energy ordered as follows $\text{BE}(\text{Ge}_{n+1}) < \text{BE}(\text{VGe}_n) < \text{BE}(\text{NbGe}_n) < \text{BE}(\text{TaGe}_n)$. Our results highlight the relative high stability of NbGe_{15} , TaGe_{15} and VGe_{14} .

Keywords: clusters; Germanium clusters; metal-doped germanium clusters; DFT; ab initio calculations;

1. Introduction

Atomic clusters containing few atoms have attracted much attention during the last decades due to their physical and chemical properties which are generally much different compared to bulk systems. The properties of these small clusters are very sensitive to the geometrical structure and they change dramatically with the composition and the number of atoms. Nanoclusters promise to be used as building blocks in advanced materials with specific properties for many applications in nanosciences and nanotechnologies [1].

Germanium is an important element in the field of semiconductor materials. Small and medium-sized germanium clusters have been intensively studied in both theoretical and experimental fields [2-4]. While pure semiconductor clusters are chemically active due to unsaturated dangling bonds, the doping with a foreign atom may lead to a significant enhancement of the stability, for example by encapsulating one or more atoms inside the cluster in order to saturate the dangling bonds[5].

During the last decades, many theoretical and experimental researches on germanium clusters Ge_n doped with different metal atoms have been reported [5-39]. Most of these works prove that the doping with a metal atom contributes to stabilize germanium cage structures and induce novel magnetic properties or special semiconductor behaviors. Among early

works, the geometries, stabilities, and electronic and magnetic properties of CuGe_n ($n = 2-13$) [6], ThGe_n ($n = 16, 18, 20$) [5], NiGe_n ($n = 1-13$) [7], WGe_n ($n = 1-17$) [8], CrGe_n ($n = 1-5$) [9], Ge_nSi_m ($m + n = 5$) [10], CoGe_n ($n = 1-13$) [11], Mo_2Ge_n ($n = 9-15$) [12], FeGe_n ($n = 9-16$) [13], $\text{Bi}_2\text{Ge}_{n-2}$ ($n = 3-8, 12$) [14], Fe_nGe_m ($n + m \leq 4$) [15], MnGe_n ($n = 2-16$) [16] clusters have been systematically investigated by using Density-Functional Theory (DFT) calculations. These studies concluded that the growth patterns depend on the nature of the doping metal atom. Particularly the charge-transfer mechanisms depend on the metal atom, and also on the size and the structure of the cluster. Most of the clusters prefer structures with a high-spin ground state and a large magnetic moment, for instance CrGe_n [9]. The magnetic moment of the doped clusters mainly originates from the $3d$ unpaired electrons of the substituted doping atom [11-13]. No quenching of the magnetic moment of the doping transition metal atom has been found for Co, Fe and Mn doped small germanium clusters [11, 13, 16]. The magnetic coupling between iron and germanium atoms is generally antiferromagnetic-like and the charge transfer mainly occurs from Fe to Ge atoms in Fe_nGe_m clusters [15]. In the case of Mo_2Ge_n clusters, the Mo_2 dimer moves to the center of the cage and becomes completely encapsulated into the germanium clusters as the size increases [12].

In the last decade, many new investigations on metal atom doped germanium clusters have been reported. Physical properties of TMGe_n ($\text{TM} = \text{Sc}, \text{Ti}, \text{V}; n = 14-20$) [17], NiGe_n ($n = 1-20$) [18] and CuGe_n ($n = 1-20$) clusters [19] have been investigated by bandyopadhyay and coworkers. They found that a relative high stability of TM ($\text{Sc}, \text{Ti}, \text{V}$) doped germanium clusters may be related to the formation of a filled shell free-electron gas inside the germanium cage or to the geometric effects [17]. The growth behavior, electronic and magnetic properties of TMGe_n ($\text{TM} = \text{Mn}, \text{Co}, \text{Ni}, \text{Cr}; n = 1-13$) [20, 21], ($\text{TM} = \text{Ti}, \text{Zr}$ and $\text{Hf}; n = 1-20$) [22], Sn_mGe_n ($m + n \leq 5$) [23], AlGe_n ($n = 1-9$) [24] clusters have been also investigated. It was found that the magnetic behavior of TMGe_n ($\text{TM} = \text{Mn}, \text{Co}, \text{Ni}, \text{Cr}$) clusters is due to TM doping atoms and the magnetic moment is mainly localized at the TM site and neighbouring Ge atoms [20, 21]. The Al atom does not enhance the binding energy of germanium clusters in AlGe_n [24]. The CoGe_n^- ($n = 2-11$) [25] and VGe_n^- ($n = 3-12$) [26] clusters were also investigated with anion photoelectron spectroscopy and density functional theory calculations. Both experiment and calculation show an electron transfer from the Ge_n framework to the metal atom and a minimization of the magnetic moment that is related to the structural evolution from exo- to endohedrals.

Recently, a study of the multicharged ruthenium-doped germanium clusters RuGe_n^q ($n = 2-12, q = -2, -3$) has highlighted an inherent tendency of formation of endohedral cage [27] and the decreasing of the transfer of negative charge from the Ge_n framework to the doping atom with the increasing ionization. A DFT investigation of the electronic and magnetic properties of medium-sized CrGe_n ($15 \leq n \leq 29$) [28] has shown that the encapsulation of Cr atoms within Ge_n clusters leads to stable CrGe_n cages. The atomic and electronic structure of both neutral and negatively charged ZrGe_n ($n = 1-21$) and AuGe_n ($n = 2-12$) clusters have been studied using *ab initio* calculations and compared to the experimental photoelectron spectroscopy [29,30]. For AuGe_n , the metal atom prefers the peripheral position for clusters for $n < 11$ and the endohedral one for larger n .

Very recently, extensive studies [31, 32] of the equilibrium geometries and electronic and magnetic properties of vanadium and coinage metal doped germanium clusters, MGe_n ($\text{M} = \text{V}, \text{Cu}, \text{Ag}, \text{Au}; n = 1-19$), have showed that the most stable clusters (which correspond to an endohedral cage-like structure for VGe_{14} and CuGe_{10} , or a hollow cage for AgGe_{15} and

AuGe₁₅) present a peculiar electronic structure in which the valence electrons of the doping atom and Ge atoms are delocalized and exhibit a shell structure associated with the quasi-spherical geometry. Some properties of the very small clusters VGe_n^{-/0} (n = 1-7) and CoGe_n^{-/0} (n = 1-3) have been investigated with the DFT and multi-configurational CASSCF/CASPT2 methods [33, 34, 35]; calculated vertical and adiabatic detachment energies were in agreement with the experimental results. A joint photoelectron spectroscopy and DFT study [36-37] of Fe₂Ge_n^{-/0} (n = 3-12) and Cr₂Ge_n⁻ (n = 3-14) has suggested a high potential to change the magnetic behavior of germanium clusters by introducing transition-metal dopant atoms. The atomic structures of anionic clusters ZrGe_n⁻ (n = 8-20) have been explored by DFT calculations and compared to the available experimental electronic spectra [39]; the doped anion clusters present an exohedral structure for smaller sizes while the formation of endohedral cages is favored for n ≥ 12.

Our previous work on VGe_n clusters [31] has enhanced the high stability of VGe₁₄ with present an *O_h* symmetry structure and a high binding energy, high HOMO-LUMO gap, high ionization potential, small electron affinity, and large chemical hardness. That cluster presents a peculiar electronic structure in which the valence electrons of V and Ge atoms are delocalized and exhibit a shell structure associated with the quasi-spherical geometry (1S 1P 1D 2S 1F etc.). That shell model, which was well-known in metal clusters and explained by the jellium model, was not expected in metal-doped germanium clusters. In the present work we focus on NbGe_n and TaGe_n (n=1-19) clusters, since Ta, Nb, and V are in the same column of the periodic table, in order to see whether Ta and Nb-doped Ge_n clusters also present such peculiar electronic configurations. We aim to understand their structural, electronic and magnetic properties, and to characterize their evolution of the physicochemical properties as a function of the size, the shape and the composition. It is very important to determine the relation between the structural, electronic and magnetic properties since the magnetic properties depends on the electronic structure which in turn depends on the geometry. Therefore, a systematic study of these clusters is required to understand the physical and chemical properties. According to our knowledge, no investigation of the NbGe_n and TaGe_n clusters has been reported in the literature to date.

We think that this work would be useful to understand the different properties of these new systems and could provide powerful guidelines for other theoretical and experimental studies. In section 2 we give a brief description of the theoretical method and computational scheme used in this work. In Section 3, we present and compare the results and discussion of the obtained geometries, electronic and magnetic properties of the different MGe_n clusters. Finally, Section 4 summarizes the main conclusions of our results.

2. Computational methodology

First-principles calculations using the generalized gradient approximation (GGA) with the Perdew-Burke-Ernzerhof (PBE) density functional [40] were performed by using the Siesta simulation package [41]. They have been performed by using the method of norm-conserving Troullier-Martins nonlocal pseudo-potentials [42, 43]. We used a double zeta (DZ) basis set for Ge atoms and double zeta basis plus polarization orbitals (DZP) for Nb and Ta atoms. A large cubic cell of 40 Å edge length, with the periodic boundary conditions, was taken in order to avoid the interaction between the neighboring clusters. The k grid integration has been carried using the Γ point approximation. The atomic positions have been optimized without any symmetry constraints and until the residual forces are smaller than 10⁻³ eV/Å.

A complete exploration of geometrical configurations is impossible. In the present work, a considerable number of possible initial structures (about 5-6 for very small clusters and about 50 for larger ones) were considered for each size. All isomeric structures obtained by substituting one Ge atom with a metal atom on different sites of the lowest-energy configuration of pure Ge_{n+1} clusters [31] have been considered. We have also tested structures already published for MGe_n ($\text{M}=\text{V}, \text{Ag}, \text{Cu}, \text{Au}, \text{Ni}, \text{Co}, \text{Cr}, \text{etc.}$) [18, 20, 21, 24, 31, 32], and try to build MGe_n by adding one Ge atom to MGe_{n-1} . Several spin-multiplicity states were tested.

The stability of Nb and Ta doped germanium clusters can be studied by the calculation of the binding energy, the energy difference between the Highest Occupied Molecular Orbital (HOMO) and the Lowest Unoccupied Molecular Orbital (LUMO), and the second-order energy difference. The binding energies per atom of MGe_n ($\text{M}=\text{Nb}, \text{Ta}$) are defined by the following formula:

$$E_b(\text{MGe}_n)(\text{eV/atom}) = (n E(\text{Ge}) + E(\text{M}) - E(\text{MGe}_n))/(n+1) \quad (1)$$

where $E(\text{Ge})$ is the total energy of free Ge atom, $E(\text{M})$ is the total energy of free M atom and $E(\text{MGe}_n)$ is the total energy of the MGe_n cluster. The HOMO-LUMO gap is calculated from the energy of the orbitals:

$$\Delta E (\text{eV}) = E(\text{LUMO}) - E(\text{HOMO}). \quad (2)$$

The second-order energy difference for the ground-state MGe_n ($\text{M}=\text{Nb}, \text{Ta}$) clusters can be calculated by:

$$\Delta_2 E = E(\text{MGe}_{n+1}) + E(\text{MGe}_{n-1}) - 2 E(\text{MGe}_n) \quad (3)$$

where E is the total energy of the most stable structure for each species.

At the present level of calculation, the bond lengths of Ge_2 was found to be 2.450 Å, in good agreement with the estimated experimental data of 2.44 Å [44]. The binding energy per atom, calculated to be 1.44 eV which is also in good agreement with both theoretical [6, 9, 21] and experimental (~1.35 eV) [45] data.

3. Results and discussions

3-1. Structural properties

Using the computation scheme described above, we have explored a number of isomers and determined the putative lowest-energy structure of all MGe_n ($\text{M}=\text{Nb}, \text{Ta}$ and $n=1-19$) clusters. The lowest-energy structure and some of their representative low-lying isomers are shown in Figures 1 and 2 for NbGe_n and TaGe_n respectively. The symmetry group, binding energy E_b (eV/atom), HOMO-LUMO gap ΔE (eV), the total spin magnetic moment, the vertical ionization potential VIP (eV), the vertical electron affinity VEA (eV), the chemical hardness η (eV) and the average Ge-Ge and M-Ge bond lengths are summarized in Table 1 and Table 2 for all NbGe_n and TaGe_n clusters respectively. The most stable

structures of NbGe_n clusters are generally similar to those of TaGe_n clusters. In the detailed description below, we compare the present results to those recently obtained for V-doped Ge_n [31].

The electronic ground state of NbGe and TaGe dimers is a sextet with a bond length of 2.592 and 2.603 Å respectively. The binding energy (per atom) calculated to be 1.230 and 1.358 eV for NbGe and TaGe respectively, is smaller than that of Ge₂ dimer (1.446 eV) [31]. The binding energy of TaGe is 0.160 and 0.128 eV larger than that of VGe and NbGe. For the trimers NbGe₂ and TaGe₂, the triangular structure with C_{2v} symmetry is found to be the lowest-energy structure. This ground-state structure was also obtained in the case of Ge₃ and VGe₂ trimers. NbGe₃ is a quartet and its structure is a C_{3v} pyramid similar to that reported for VGe₃, but the lowest-energy of TaGe₃ is a doublet and presents only a C_s symmetry since Ge₃ triangle is no longer equilateral but becomes isosceles with two Ge-Ge bonds are enlarged from 2.72 to 2.90 Å. The planar structure with C_{2v} symmetry which is reported as the ground state structure for Ge₄ is found to be less stable. The global minimum structure of the pentamers NbGe₄ and TaGe₄ is found to be a distorted rhombus with C_s symmetry which is in agreement with the ground state structure of VGe₄. The Ta-Ge average bond distance is respectively 0.070 and 0.081 Å higher than Nb-Ge and V-Ge average bond lengths. For the hexamers NbGe₅ and TaGe₅, the square bipyramidal geometry with C_{4v} symmetry is found to be the lowest-energy structure. The obtained ground state structure is more regular than that obtained for VGe₅ which has only a C_{2v} symmetry [31]. The Nb-Ge and Ta-Ge bond lengths are 2.631 Å and 2.658 Å respectively. The most stable isomer of the heptamers NbGe₆ and TaGe₆ is a bicapped pentagonal geometry with C_s symmetry. The similar structure is found for the most stable isomers Ge₇ and VGe₆ with D_{5h} and C_{2v} point group symmetry respectively. For NbGe₇ and TaGe₇ clusters, the capped pentagonal bipyramid structure with C_s symmetry is found to be the ground state structure, similar to the ground state of Ge₈ and VGe₇. The lowest-energy structure of the NbGe₈ and TaGe₈ clusters is a tricapped trigonal prism geometry with a C_{2v} symmetry. The same symmetry is obtained in the case of VGe₈ with the binding energy of 2.966 eV/atom [31]. In the case of NbGe₉ and TaGe₉ clusters, the lowest-energy structure can be viewed as a pentagonal bipyramid face-capped by three Ge atoms, it is much different from VGe₉ which is formed by two distorted hexagonal prisms and a C_s symmetry. We can note that the binding energies of NbGe₉ and TaGe₉ are respectively 0.044 and 0.055 eV/atom smaller comparatively to the ground state structure of NbGe₈ and TaGe₈ clusters which may be related to the loss of symmetry. For NbGe₁₀ and TaGe₁₀ clusters, the metal atom is partially encapsulated in an opened cage with C_2 symmetry. This structure has been already found as the most stable isomer for VGe₁₀ cluster [31]. The average Ge-Ge and Ta-Ge bond lengths are 0.005 Å and 0.109 Å larger than Nb-Ge and V-Ge average bond lengths respectively. For the size $n = 11$, the optimized ground state structure of Nb-doped Ge_n clusters differ slightly from that of Ta-doped Ge_n. A distorted superposition of two rhombic structures with C_2 symmetry is found to be the lowest-energy isomer of TaGe₁₁ which is consistent with the ground structure of VGe₁₁. The Nb atom occupies a more central position and tends to form more bonds with germanium atoms, therefore NbGe₁₁ seems to be more an opened cage with ten Nb-Ge bonding. Its binding energy is 0.015 eV/atom smaller than that of NbGe₁₀.

Starting from $n = 12$, the endohedral structures where the Nb or Ta metal atom is highly coordinated and totally encapsulated inside a germanium cage appear, while the exohedral structures lie much higher than the endohedrals. Similar to VGe₁₂ clusters, a perfect hexagonal prism isomer having a D_{3d} symmetry with the Nb or Ta doping atom encapsulated

into a caged Ge_{12} framework is found to be the ground-state isomer in NbGe_{12} and TaGe_{12} clusters. These ground-state isomers have the shortest Ge-Ge bond lengths of 2.593 and 2.594 Å, respectively. However, a second isomer, labeled $\text{NbGe}_{12}\text{-b}$ and $\text{TaGe}_{12}\text{-b}$ respectively, with C_s symmetry lies only 0.003 and 0.004 eV/atom above the D_{3d} -symmetry isomer at the present level of calculation. Then isomers are quasi-degenerate. The second structures is somewhat similar to the lowest-energy one but one Ge atom comes to cap a hexagonal face Ge_6 and thus closing the cage on one side. In NbGe_{13} and TaGe_{13} clusters, the lowest-energy isomer is a hexagonal prism capped bowl kind of structure with Nb and Ta metal atoms inside the cage with C_{3v} symmetry. In the case of VGe_{13} , the most stable structure is also a capped hexagonal prism but with the high C_{6v} symmetry. The average Nb-Ge and Ta-Ge bond lengths are 0.192 Å and 0.193 Å larger than the average Ge-Ge bond length. The most stable isomer of TaGe_{13} is less stable than TaGe_{12} by only 0.002 eV/atom. The lowest-energy structures of NbGe_{14} and TaGe_{14} are the perfect sealed cage structures with C_2 symmetry, the metal doping atom is highly coordinated and located at the center of the cage. The endohedral structures $\text{NbGe}_{14}\text{-b}$ and $\text{TaGe}_{14}\text{-b}$ with high O_h symmetry, which have been reported as the ground state structure in VGe_{14} [31], are found at 0.069 and 0.017 eV/atom respectively above the lowest-energy isomer. They do not compete for the ground state, but they are metastable since all frequencies are calculated to be positive.

The most stable structures for the sizes $n = 15, 16$ and 17 obviously derive from the ground-state for $n = 14$ by adding one Ge atom, two Ge atoms and three Ge atoms, respectively. The lowest-energy structure of the NbGe_{15} and TaGe_{15} clusters has a C_{2v} symmetry with Nb-Ge and Ta-Ge bond-lengths of 2.987 and 2.990 Å, respectively. This endohedral structure has already been found as the lowest-energy isomer in the case of VGe_{15} . The ground state geometries obtained from $n = 16$ to 18 for each species are all less stable than the corresponding TaGe_{15} and NbGe_{15} clusters. The lowest-energy structure of NbGe_{16} is $\text{NbGe}_{16}\text{-a}$ with C_1 point group symmetry and the lowest-energy isomer of TaGe_{16} is $\text{TaGe}_{16}\text{-a}$ with C_s symmetry and the largest Ta-Ge average bond length of 3.019 Å. However, the $\text{NbGe}_{16}\text{-b}$ cluster with C_s symmetry which has the largest Nb-Ge average bond length of 3.016 Å is found to lie only 0.002 eV/atom above the lowest-energy isomer of NbGe_{16} . The lowest-energy isomer of VGe_{16} is an irregular cage-like structure with C_1 symmetry. The endohedral ground state geometries for $n = 17$ belong the C_1 symmetry group with average Nb-Ge and Ta-Ge bond distances of 2.948 Å and 2.961 Å respectively. A prolate structure has been previously predicted to be the ground state for VGe_{17} clusters. The lowest-energy structures of NbGe_{18} and TaGe_{18} are the irregular cage-like with respectively C_s and C_1 symmetry and the doping atom completely encapsulated at the center of the cage. A prolate-like cage symmetry structure with C_1 was obtained as the lowest-energy isomer of VGe_{18} clusters. For NbGe_{19} , two structures ($\text{NbGe}_{19}\text{-a}$ and $\text{NbGe}_{19}\text{-b}$) compete for the lowest-energy isomer. In the lowest-energy isomer ($\text{NbGe}_{19}\text{-a}$), the Ge-Ge average bond length is much larger (2.806 vs 2.730 Å) and the Nb-Ge average bond length is much shorter (2.950 vs 2.978 Å). The lowest-energy isomer for TaGe_{19} cluster has the shortest Ge-Ge and the largest Ta-Ge average bond lengths of 2.696 and 3.009 Å respectively. The binding energies of NbGe_{19} and TaGe_{19} are respectively 0.010 and 0.029 eV/atom larger comparatively to the ground state structures of NbGe_{18} and TaGe_{18} clusters.

3-2. Electronic properties

Binding energy

By using the formula (1), we calculate the binding energies for all the $M\text{Ge}_n$ ($M = \text{Nb}$, Ta and $n=1-19$) clusters. The obtained results are listed in Table 1 and 2 and their evolution with the cluster size are plotted in Figure 3. The size dependence shows an increasing of the binding energies for all pure and M -doped ($M = \text{V}$, Nb , Ta) germanium clusters: a dramatic increasing of the binding energy is observed for the very small size ($n < 8$), then the binding energy increases more smoothly. We also observe that the binding energy of TaGe_n clusters are generally the largest comparatively to the other species followed by the binding energies of NbGe_n , then the binding energy of VGe_n and finally the binding energy of pure corresponding germanium clusters Ge_{n+1} . Doping with Nb (resp. V) enhances the stability of Ge_n clusters for $n \geq 7$ (resp. 8), while doping with Ta has always a stabilizing character. The nature of bonding between the M atom and neighboring Ge atoms can explain this difference in the evolution of binding energy as a function of the doping atom species. Ta presents a stronger covalent character than Nb which in turn is more covalent than the vanadium. Thus the binding energy of the dimer TaGe , NbGe , VGe is calculated at 1.358, 1.230, 1.198 eV/atom respectively.

Additionally, the average binding energy curve presents a local maximum value at $n = 8$ and 15 for both Nb - and Ta -doped Ge_n implying that the clusters are relatively more stable than their neighbors. Actually, the local maximum for $n=8$ is due to the relative low stability of metal doped Ge_9 , likely related to the loss of symmetry and a relatively low coordination of the metal atom. Then, the largest value of binding energy is observed in the case of TaGe_{15} and NbGe_{15} .

Another parameter which can reflect the stability of the small clusters is the second-order energy difference (Δ_2E) defined by the relation (3). It is physically related to the experimental relative abundance of the corresponding element in the mass spectroscopy experiments. The calculated second-order energy difference of the lowest-energy isomers of different species studied in this work are shown in Figure 4. As we can see, the pronounced positive values of Δ_2E are observed for Nb-TaGe_8 and $\text{V-Nb-TaGe}_{10,12,15}$ indicating that these clusters may have special stabilities.

Nb - and Ta -doped Ge_{15} present a relative high stability, while the most stable structure for V -doped Ge_n was obtained for the size $n=14$ for which a perfect O_h -symmetry cage was obtained [31]. The atomic radius of Nb and Ta being slightly higher than that of V , it was probable that the optimal cage would be obtained for n slightly greater than 14. The high stability of VGe_{14} was associated to a peculiar electronic structure since the valence electrons of both Ge and V were found to be delocalized and exhibit a shell structure associated to the quasi-spherical geometry, with the following occupation: $1S^2 1P^6 1D^{10} 2S^2 1F^{14} 3S^2 2P^6 1G^{15} 2D^4$ [31]. In Figure 5, we show the density of states (DOS) and the Kohn-Sham orbitals calculated at PBE/cc-pvtz level with the software Gaussian09 [46] in the case of TaGe_{15} . The 3s and 3p valence electrons of Ge and the five s and d valence electrons of Ta exhibit a shell structure associated to the somewhat spherical structure. We can easily distinguish the character of the orbitals, though there are some little deviations from a perfect sphere due to explicit location of atoms and the C_{2v} symmetry instead of K_h (the symmetry of the atom). The 65 valence electrons of the cluster are organized with the following occupations: $1S^2 1P^6 1D^{10} 1F^{14} 2S^2 2P^6 1G^{18} 2D^6 1H^1$. The number of electrons does not exactly fit with shell

closings numbers, but the pooling of electrons and the organization in shell contributes to the high stability of the cluster. Similar results are found for NbGe₁₅.

HOMO-LUMO gaps

The HOMO-LUMO gap is one important parameter that characterizes the chemical stability of the small clusters. A small value of HOMO-LUMO energy gap corresponds to a high chemical reactivity while a large value is related to an enhanced chemical stability. The size dependence of the calculated HOMO-LUMO gaps for all the ground-state structures is shown in Figure 6. An oscillating behavior is observed in the evolution of HOMO-LUMO gaps as the size increases. The HOMO-LUMO gaps of metal doped germanium clusters are generally smaller than that of the corresponding pure germanium cluster which means the doping metal atom enhances the reactivity of relevant clusters and favors the apparition of the metallic behavior, especially for smallest clusters where the metal is located at an exohedral position. For the cage-like structures, VGe₁₂₋₁₄ present clearly a higher gap than Nb- and Ta-doped clusters, indicating that these clusters may have a relative low reactivity.

VIP, VEA and η

The vertical ionization potential (VIP) and the vertical electron affinity (VEA) are two important parameters which can determine the chemical stability behavior of the small clusters. The VIP is defined by the energy difference between the cationic $E(MGe_n)^+$ and neutral clusters calculated at the equilibrium geometry of the neutral MGe_n cluster, the VEA is the energy difference between the neutral $E(MGe_n)$ and the anionic clusters calculated at the geometry of the neutral cluster, as given in the following equations:

$$VIP = E(MGe_n)^+ - E(MGe_n) \quad (4)$$

$$VEA = E(MGe_n) - E(MGe_n)^- \quad (5)$$

Results are reported in Tables 1 and 2 for NbGe_n and TaGe_n clusters and their evolution as a function of the cluster size are plotted in Figures 7 and 8 respectively. VIP values are between 6 and 7.3 eV. An oscillating behavior is observed in the evolution of VIP for Nb and Ta doped germanium clusters. Local maxima are found for $n = 10, 12$, and 14 . The effect of the doping is clearly visible for very small n and for $n > 15$ where the metal atom induces a significant decrease of the VIP, but for $n = 10-15$ the effect of the dopant is weak because the metal is highly coordinated in a compact cage.

In Figure 8 we see that the VEA is sensitive to the size and shows an increasing evolution with the increasing cluster size which indicates that the clusters with large size will capture an electron more easily. The metal doped Ge_n have a stronger VEA than the corresponding Ge_{n+1} cluster, except for $n=14$.

The chemical hardness (η) is defined as the difference between the vertical ionization potential and electronic affinity as:

$$\eta = VIP - VEA \quad (6)$$

It is considered as a good parameter to understand the chemical stability of small clusters. The clusters with high values of hardness are less reactive [47]. The size dependence of the chemical hardness for all MGe_n clusters are shown in Figure 9. The value of η presents a roughly decreasing evolution with increasing size for both $NbGe_n$ and $TaGe_n$ without peculiar values, except a local minima for $n=11$, already visible for Ge_{n+1} . In contrast the size dependence of η for V-doped Ge_n clusters highlights the size $n = 14$ which presents the highest stability.

3-3. Magnetic properties

The magnetic properties of the small Nb-Ta- Ge_n clusters is discussed on the basis of the total spin magnetic moment (TSMM) as the difference between the total Mulliken charge populations in spin up and spin down electrons. The calculated TSMM for all isomers are reported in Tables 1 and 2. As we can see, the TSMM is generally equal to $1 \mu_B$ except for the very small clusters where $\mu = 5\mu_B$ for $NbGe$ and $3\mu_B$ for $NbGe_{2,3}$. For $TaGe_n$ clusters the TSMM is 5 and $3 \mu_B$ for $TaGe$ and $TaGe_2$ respectively. Partial densities of states in Figure 10 in order to examine the contribution of different valence orbitals (s, p and d) of the components. The spin up densities is plotted as positive and the spin down as negative. For $NbGe_1$ which have the high value of $5 \mu_B$, the TSMM is due to the contribution of two species by 4s and 4p orbitals of Ge and the 4d orbital of Nb atom. However, in the $NbGe_2$ and $NbGe_3$ clusters, the TSMM is mainly due to the 4d orbitals of Niobium atom. In the case of $TaGe_n$ clusters all of the valence orbitals participate to the TSMM with a large contribution of the 5d orbital of the Tantalum atom. In all of other clusters the TSMM of the doping atom is quenched, it should be due to orbital hybridizations and charge transfers between M and surrounding Ge atoms.

4. Conclusion

First-principles DFT calculations have been performed to study the structures, stabilities, electronic and magnetic properties of Nb and Ta atoms doped germanium clusters, MGe_n ($M = Nb, Ta$ and $n=1-19$). The growth pattern for MGe_n ($M = Nb, Ta$) shows that the M atom occupies a peripheral position for very small clusters ($n < 10$), while for $n \geq 10$ the endohedral structures in which the germanium cage encapsulates the metal atom are strongly favored. The binding energy increases with the increasing cluster size for all MGe_n ($M = Nb, Ta$) clusters and the M atoms enhances the stabilities of doped clusters comparatively to that of the corresponding pure germanium cluster. This stability is related to the pooling of valence electrons from both germanium and metal atoms. $TaGe_n$ clusters present the highest stabilities. $TaGe_{15}$ and $NbGe_{15}$ are predicted to present a relative high stability. All of the isomers of Nb and Ta doped germanium clusters have the total spin magnetic moment equal to $1 \mu_B$ except for the very small size ($n \leq 3$ for $NbGe_n$ and $n \leq 2$ for $TaGe_n$). PDOS analysis reveals the high total spin magnetic moment for the very small clusters is due to the contribution of the different valence orbitals with a large domination of 4d orbitals of Nb and 5d orbitals of Ta atoms. Finally, we hope that these results might be helpful for further theoretical and experimental investigations on future nano-sized materials.

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Table 1. Symmetry group, binding energy E_b (eV/atom), HOMO-LUMO gap ΔE (eV), vertical ionization potential VIP (eV), vertical electron affinity VEA (eV), chemical hardness η (eV) and average bond length $a_{\text{Ge-Ge}}$ and $a_{\text{Ge-Nb}}$ (Å) for NbGe_n clusters. Data for the lowest-energy isomer are in bold.

n	Label-symmetry	E_b (eV/atom)	ΔE (eV)	μ (μ_B)	VIP (eV)	VEA (eV)	η (eV)	$a_{\text{Ge-Ge}}$ (Å)	$a_{\text{Ge-Nb}}$ (Å)
1	a- $C_{\infty v}$	1.230	0.704	5	6.645	0.697	5.948	-	2.592
2	a- C_{2v}	2.198	1.060	3	7.083	0.881	6.202	2.432	2.596
	b- $D_{\infty h}$	1.375	1.177	9	6.712	1.925	4.787	-	2.728
3	a- C_{3v}	2.370	0.406	3	6.242	1.551	4.691	2.682	2.695
	b- C_{2v}	2.357	0.468	3	6.551	1.599	4.952	2.425	2.731
	c- C_s	2.024	0.261	5	6.902	2.057	4.845	2.466	2.627
4	a- C_s	2.613	0.424	1	6.786	1.930	4.856	2.813	2.611
	b- C_2	2.573	0.295	3	6.556	1.776	4.780	2.660	2.794
	c- C_2	2.500	0.507	3	6.386	1.600	4.786	2.548	2.598
	d- C_{2v}	2.483	0.200	3	6.660	2.276	4.384	2.486	2.725
	e- D_{2h}	2.300	0.401	1	6.932	2.356	4.576	2.593	2.553
5	a- C_{4v}	2.829	0.901	1	7.269	2.262	5.007	2.770	2.631
	b- C_1	2.750	0.721	3	6.658	2.073	4.585	2.703	2.685
	c- C_1	2.687	0.527	1	6.896	2.144	4.752	2.680	2.649
6	a- C_s	2.941	1.014	1	7.236	2.231	5.005	2.737	2.774
	b- C_s	2.856	0.366	3	6.273	1.802	4.471	2.749	2.812
	c- C_1	2.796	0.630	3	6.397	1.695	4.702	2.744	2.804
	d- C_s	2.754	0.929	3	6.764	2.396	4.367	2.705	2.757
	e- C_{2v}	2.704	0.334	3	6.849	2.557	4.292	2.759	2.661
7	a- C_s	3.026	0.497	1	6.956	2.634	4.322	2.728	2.803
	b- C_s	2.992	0.837	1	6.711	1.773	4.938	2.724	2.752
	c- C_s	2.986	0.732	1	6.885	2.382	4.503	2.746	2.705
	d- C_{2v}	2.924	0.737	3	6.356	1.899	4.457	2.740	2.641
	e- C_1	2.846	0.602	1	6.554	2.442	4.112	2.657	2.811
	f- C_s	2.805	0.607	1	6.304	2.081	4.223	2.682	2.626
8	a- C_{2v}	3.113	0.852	1	6.811	2.582	4.229	2.759	2.834
	b- C_1	3.022	0.908	3	6.890	2.495	4.395	2.648	2.805
	c- C_s	3.008	0.627	1	6.454	2.294	4.160	2.737	2.767
	d- C_1	2.902	0.357	1	6.344	2.279	4.065	2.714	2.832
	e- O_h	2.776	0.460	3	7.041	2.854	4.187	3.001	2.599
9	a- C_1	3.069	0.551	1	6.770	2.535	4.235	2.745	2.860
	b- C_1	3.043	0.884	1	6.739	2.328	4.411	2.718	2.773
	c- C_2	3.027	0.446	1	6.975	2.954	4.021	2.700	2.770
	d- C_{2v}	3.018	0.418	1	6.266	2.105	4.161	2.747	2.811
	e- C_s	2.977	0.476	1	6.573	2.455	4.118	2.757	2.755
	f- C_1	2.971	0.723	1	6.401	2.113	4.288	2.762	2.784
	g- C_s	2.931	0.620	3	6.535	2.610	3.925	2.739	2.773
10	a- C_2	3.164	0.529	1	7.016	2.874	4.142	2.846	2.870
	b- C_1	3.105	0.517	1	6.603	2.554	4.049	2.774	2.820
	c- C_s	3.042	0.663	1	6.550	2.364	4.186	2.796	2.725
	d- C_1	2.964	0.540	3	6.136	2.299	3.837	2.810	2.846
11	a- C_{2v}	3.149	0.463	1	6.950	3.012	3.938	2.805	2.806
	b- C_s	3.112	0.595	1	6.745	2.779	3.966	2.754	2.865

	c- C _s	3.064	0.799	1	6.324	1.734	4.590	2.714	2.729
	d- C ₁	3.035	0.647	1	6.386	2.037	4.349	2.791	2.748
	e- C _s	2.999	0.471	3	6.293	2.503	3.790	2.788	2.765
12	a- D_{3d}	3.243	0.248	1	7.128	2.796	4.332	2.593	2.895
	b- C _s	3.240	0.438	1	7.127	3.276	3.851	2.778	2.908
	c- C ₁	3.177	0.601	1	6.774	3.012	3.762	2.744	2.885
	d- C ₁	3.167	0.636	1	6.633	2.900	3.733	2.764	2.854
13	a- C_{3v}	3.250	0.155	1	6.848	2.492	4.356	2.724	2.916
	b- C ₁	3.177	0.684	1	6.528	2.860	3.668	2.797	2.881
	c- C ₁	3.177	0.603	1	6.548	2.940	3.608	2.697	2.842
	d- C ₁	3.092	0.757	1	6.330	2.315	4.015	2.775	2.820
	e- C _{4v}	3.012	0.496	1	6.504	2.713	3.791	2.764	2.663
14	a- C₂	3.309	0.354	1	7.095	2.973	4.122	2.769	2.970
	b- O _h	3.240	1.071	1	7.055	3.024	4.031	2.702	2.929
	c- C _{2v}	3.082	0.500	3	6.552	2.871	3.681	2.809	2.726
	d- C ₁	3.033	0.782	1	6.052	2.419	3.633	2.712	2.545
15	a- C_{2v}	3.334	0.678	1	6.444	2.909	3.535	2.747	2.987
	b- C ₁	3.209	0.797	1	6.467	2.700	3.767	2.743	2.959
	c- C ₁	3.135	0.694	1	6.321	2.618	3.703	2.735	2.918
	d- C ₁	3.032	0.650	3	6.330	2.716	3.614	2.787	2.773
16	a- C₁	3.304	0.708	1	6.315	2.847	3.468	2.750	3.008
	b- C _s	3.302	0.661	1	6.159	2.637	3.522	2.749	3.016
	c- C ₁	3.258	0.667	1	6.493	2.877	3.616	2.701	2.938
	d- C ₁	3.204	0.728	1	6.450	2.946	3.504	2.811	2.917
17	a- C₁	3.265	0.558	1	6.012	2.580	3.432	2.762	2.948
	b- C ₁	3.204	0.435	1	6.334	2.938	3.396	2.769	2.994
	c- C ₁	3.166	0.574	1	6.301	2.943	3.358	2.753	2.853
	d- C ₁	3.152	0.401	1	6.366	3.113	3.253	2.759	2.752
	e- C _s	3.059	0.436	1	6.170	2.546	3.624	2.713	2.781
18	a- C_s	3.245	0.312	1	6.101	2.796	3.305	2.684	2.973
	b- C ₁	3.226	0.667	1	6.382	3.003	3.379	2.751	2.900
	c- C ₁	3.218	0.441	1	6.048	2.791	3.257	2.696	2.979
	d- C _s	3.191	0.698	1	6.221	2.943	3.278	2.673	2.906
19	a- C₁	3.255	0.669	1	6.299	3.054	3.245	2.806	2.950
	b- C ₁	3.254	0.629	1	6.564	3.067	3.497	2.730	2.978
	c- C ₁	3.251	0.580	1	6.557	3.019	3.538	2.749	2.970
	d- C ₁	3.200	0.474	1	6.127	2.910	3.217	2.726	2.953
	e- C ₁	3.185	0.733	1	6.225	2.882	3.343	2.717	2.872

Table 2. Symmetry group, binding energy E_b (eV/atom), HOMO-LUMO gap ΔE (eV), vertical ionization potential VIP (eV), vertical electron affinity VEA (eV), chemical hardness η (eV) and average bond length $a_{\text{Ge-Ge}}$ and $a_{\text{Ge-Ta}}$ (Å) for TaGe_n clusters. Data for the lowest-energy isomer are in bold.

n	Label-symmetry	E_b (eV/atom)	ΔE (eV)	μ (μ_B)	VIP (eV)	VEA (eV)	η (eV)	$a_{\text{Ge-Ge}}$ (Å)	$a_{\text{Ge-Ta}}$ (Å)
1	a- $C_{\infty v}$	1.358	0.783	5	6.699	0.561	6.138	-	2.603
2	a- C_{2v}	2.329	1.376	3	7.005	1.007	5.998	2.453	2.579
	b- $D_{\infty h}$	1.537	0.265	3	6.481	1.809	4.672	-	2.489
3	a- C_s	2.646	0.345	1	7.185	1.716	5.469	2.781	2.523
	b- C_{2v}	2.487	0.485	3	6.633	1.557	5.076	2.428	2.723
	c- C_{2v}	2.178	0.149	3	6.702	2.071	4.631	2.500	2.558
4	a- C_s	2.792	0.649	1	6.894	1.831	5.062	2.745	2.681
	b- C_{2v}	2.709	0.604	1	6.913	1.859	5.054	2.494	2.674
	c- C_1	2.626	0.476	3	6.436	1.746	4.690	2.693	2.703
	d- D_{2h}	2.450	0.368	1	6.903	2.317	4.586	2.599	2.569
5	a- C_{4v}	2.964	0.826	1	7.305	2.280	5.025	2.774	2.658
	b- C_1	2.905	0.443	1	6.880	2.208	4.672	2.669	2.767
	c- C_s	2.811	0.645	1	6.688	2.154	4.534	2.786	2.668
6	a- C_s	3.054	0.898	1	7.297	2.400	4.897	2.741	2.770
	b- C_s	3.025	0.575	1	6.991	2.403	4.588	2.547	2.774
	c- C_{2v}	2.966	0.810	1	6.604	2.112	4.492	2.745	2.740
	d- C_s	2.918	0.463	3	6.199	1.644	4.555	2.760	2.794
	e- C_1	2.894	0.607	3	6.829	2.640	4.189	2.706	2.736
7	a- C_s	3.133	0.427	1	7.018	2.679	4.339	2.731	2.801
	b- C_s	3.097	0.703	1	6.764	2.315	4.449	2.748	2.719
	c- C_s	3.097	0.701	1	6.762	2.314	4.448	2.748	2.719
	d- C_{2v}	3.019	0.666	1	6.126	1.701	4.425	2.737	2.652
	e- C_1	2.906	0.549	1	6.217	2.027	4.190	2.684	2.621
8	a- C_{2v}	3.221	0.841	1	6.787	2.570	4.217	2.763	2.839
	b- C_s	3.110	0.705	1	6.429	2.220	4.209	2.742	2.761
	c- C_2	3.017	0.402	3	6.984	2.855	4.129	2.819	2.826
	d- O_h	2.879	0.451	3	7.018	2.851	4.167	3.020	2.615
9	a- C_1	3.166	0.509	1	6.748	2.537	4.211	2.747	2.862
	b- C_1	3.130	0.784	1	6.663	2.404	4.259	2.722	2.773
	c- C_2	3.125	0.439	1	6.954	2.968	3.986	2.701	2.779
	d- C_{2v}	3.103	0.293	1	6.155	2.257	3.898	2.767	2.835
	e- C_1	3.044	0.629	1	6.230	2.029	4.201	2.766	2.777
	f- C_s	3.020	0.909	3	6.323	2.158	4.165	2.811	2.720
10	a- C_2	3.256	0.494	1	7.084	2.877	4.207	2.833	2.875
	b- C_s	3.114	0.596	1	6.617	2.440	4.066	2.798	2.724
	c- C_s	3.095	0.360	1	6.506	2.684	3.933	2.768	2.680
	d- C_1	3.090	0.431	1	6.213	2.348	3.865	2.720	2.772
11	a- C_2	3.272	0.527	1	6.875	3.071	3.804	2.715	2.904
	b- C_s	3.255	0.682	1	6.666	2.886	3.780	2.672	2.844
	c- C_s	3.172	0.570	1	6.342	2.379	3.963	2.756	2.822
	d- C_1	3.155	0.565	1	6.347	2.415	3.932	2.705	2.843
	e- C_s	3.127	0.761	1	6.150	2.034	4.116	2.719	2.733
12	a- D_{3d}	3.323	0.260	1	7.124	2.827	4.297	2.594	2.898

	b- C _s	3.319	0.420	1	7.067	3.258	3.809	2.794	2.909
	c- C ₂	3.278	0.647	1	6.740	2.977	3.763	2.736	2.846
	d- C ₁	3.247	0.645	1	6.624	2.901	3.723	2.765	2.861
	e- C ₁	3.163	0.609	1	6.361	2.668	3.693	2.783	2.804
13	a- C_{3v}	3.321	0.563	1	6.824	2.497	4.327	2.727	2.920
	a- C ₁	3.252	0.683	1	6.528	2.886	3.642	2.786	2.855
	c- C ₁	3.250	0.600	1	6.511	2.935	3.576	2.716	2.849
	d- C ₁	3.246	0.499	1	6.548	2.944	3.604	2.739	2.847
	e- C ₁	3.156	0.760	1	6.283	2.437	3.846	2.777	2.817
	f- C ₁	3.133	0.537	1	6.531	2.718	3.813	2.793	2.678
14	a- C₂	3.381	0.345	1	7.106	2.952	4.154	2.772	2.974
	b- C ₁	3.364	0.649	1	6.887	3.244	3.643	2.723	2.979
	c- O _h	3.301	0.976	3	7.016	3.102	3.914	2.707	2.932
	d- C ₁	3.087	0.686	1	5.985	2.527	3.458	2.743	2.558
15	a- C_{2v}	3.401	0.687	1	6.441	2.910	3.531	2.749	2.990
	b- C ₁	3.319	0.532	1	6.383	2.958	3.425	2.678	2.947
	c- C ₁	3.243	0.524	1	6.127	2.476	3.651	2.753	2.903
	d- C ₁	3.141	0.833	1	6.358	2.659	3.699	2.765	2.735
16	a- C_s	3.365	0.707	1	6.170	2.652	3.518	2.750	3.019
	b- C ₁	3.331	0.642	1	6.551	3.066	3.485	2.634	2.896
	c- C ₁	3.265	0.697	1	6.419	2.965	3.454	2.814	2.922
	d- C ₁	3.143	0.450	1	6.169	2.773	3.396	2.752	2.876
17	a- C₁	3.323	0.592	1	6.037	2.601	3.436	2.760	2.961
	b- C ₁	3.262	0.448	1	6.330	2.966	3.364	2.769	2.980
	c- C ₁	3.223	0.572	1	6.286	2.951	3.335	2.755	2.861
	d- C ₁	3.209	0.547	1	6.393	3.090	3.303	2.745	2.777
	e- C _s	3.107	0.365	1	6.061	2.679	3.382	2.714	2.782
18	a- C₁	3.279	0.608	1	6.357	3.037	3.320	2.741	2.905
	b- C _{2v}	3.250	0.410	1	6.134	2.909	3.225	2.616	2.948
	c- C _s	3.246	0.706	1	6.220	2.956	3.264	2.673	2.911
	d- C ₂	3.233	0.526	1	6.443	3.149	3.294	2.776	2.851
	e- C _s	3.138	0.538	1	5.967	2.584	3.383	2.810	2.657
	f- C ₁	3.114	0.531	1	6.117	2.938	3.179	2.738	2.778
19	a- C₁	3.308	0.463	1	6.500	3.161	3.339	2.696	3.009
	b- C ₁	3.265	0.523	1	6.302	2.982	3.320	2.756	3.003
	c- C ₁	3.260	0.477	1	6.328	3.036	3.292	2.707	2.992
	d- C _s	3.235	0.700	1	6.214	2.934	3.280	2.703	2.876
	e- C _s	3.223	0.271	1	6.279	3.055	3.224	2.738	2.950

Graphical Abstract

Figure 1. Lowest-energy structures and their corresponding isomers of NbGe_n (n=1-19) clusters. For each size, the lowest-energy isomer is indicated in bold.

Figure 2. Lowest-energy structures and their corresponding isomers of TaGe_n (n=1-19) clusters. For each size, the lowest-energy isomer is indicated in bold.

Figure 3. Binding energy (eV/atom) of the lowest-energy structures of MGe_n ($\text{M} = \text{V}$ [31], Nb, Ta) ($n=1-19$) clusters.

Figure 4. Second-order energy differences Δ_2E (eV) of the lowest-energy structures of MGe_n ($M= V$ [31], Nb, Ta) ($n=1-19$) clusters.

Figure 5. Density of states (DOS) of TaGe₁₅ for alpha spin electrons. For beta spin electrons, the 1H orbital is empty (doublet state). For each band, the Kohn-Sham orbitals are plotted. DOS for NbGe₁₅ is very similar.

Figure 6. HOMO-LUMO gaps (eV) of the lowest-energy structures of MGe_n ($\text{M} = \text{V}$ [31], Nb, Ta) ($n=1-19$) clusters.

Figure 7. Vertical ionization potential VIP (eV) of the lowest-energy structures of MGe_n ($\text{M} = \text{V}$ [31], Nb, Ta) ($n=1-19$) clusters.

Figure 8. Vertical electron affinity VEA (eV) of the lowest-energy structures of MGe_n ($\text{M} = \text{V}$ [31], Nb, Ta) ($n=1-19$) clusters.

Figure 9. Chemical hardness (eV) of the lowest-energy structures of MGe_n ($\text{M} = \text{V}$ [31], Nb, Ta) ($n=1-19$) clusters.

Figure 10. Projected densities of states of $\text{NbGe}_{1,2,3}$ and $\text{TaGe}_{1,2}$ clusters. The Fermi level has been taken as the origin of the energy.